Structures of $ErPO_4$, $TmPO_4$, and $YbPO_4$

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(Received 17 May 1982; accepted 5 August 1982)

Abstract. $M_r = 262 \cdot 2$, $263 \cdot 9$, and $268 \cdot 0$, respectively, tetragonal, $I4_1/amd$, zircon structure, $a = 6 \cdot 860$ (1), $6 \cdot 839$ (1) and $6 \cdot 816$ (2), $c = 6 \cdot 003$ (1), $5 \cdot 986$ (1), and $5 \cdot 966$ (2) Å, $V = 282 \cdot 5$, $280 \cdot 0$, and $277 \cdot 2$ Å³, Z = 4, $D_x = 6 \cdot 164$, $6 \cdot 259$, and $6 \cdot 421$ Mg m⁻³, Mo Ka($\lambda = 0 \cdot 71073$ Å), 292 K, R_1 and $R_2 = 0 \cdot 035$, $0 \cdot 016$, and $0 \cdot 035$ for 411, 332, and 335 reflections, respectively; F(000) = 460, 464, and 468. The lanthanoid metal atom in each compound is coordinated to eight O atoms that form two orthogonal interpenetrating tetrahedra. Each has distinct bond distances.

Introduction. The structural investigations of $ErPO_4$, $TmPO_4$, and $YbPO_4$ reported here are part of an extensive research program dealing with the practicability of employing the lanthanoid orthophosphates as fundamental hosts for the isolation of nuclear wastes. The potential for accommodating chemically diverse cations such as thorium, uranium, α -active transuranic isotopes, and fission products such as Sr and Cs, which are produced during the operation of a nuclear reactor, in the lattice of the predominantly trivalent LnPO₄ systems has been discussed elsewhere (Beall, Boatner, Mullica & Milligan, 1981). These compounds have the zircon structure (Schwarz, 1963).

Experimental. Single crystals of $ErPO_4$, $TmPO_4$, and $YbPO_4$ were grown by means of the flux technique (Feigelson, 1964). The appropriate lanthanoid oxide (*i.e.* Er_2O_3 , Tm_2O_3 , or Yb_2O_3) and PbHPO_4 were placed in a Pt crucible that was closed with a tightly fitting lid. PbHPO_4 decomposes at high temperature to form $Pb_2P_2O_7$ which then serves as the flux matrix. Lead pyrophosphate reacts with the lanthanoid oxide to form the appropriate lanthanoid orthophosphate, and, after soaking the mixture for approximately 16 h at 1633 K, the crucible was slowly cooled to 1173 K at a rate of 1 K h⁻¹. The Pt crucible was then allowed to cool directly to room temperature at which time the

 $Pb_2P_2O_7$ flux was dissolved in boiling HNO₃ in order to free the orthophosphate crystals. A more complete description of the crystal-growth process has been given elsewhere (Rappaz, Boatner & Abraham, 1980).

Single crystals of ErPO₄, TmPO₄, and YbPO₄ selected on the basis of their optical quality; cell dimensions from 25 centered reflections in the range $1.5 < \theta < 25^{\circ}$ for ErPO₄, TmPO₄, and YbPO₄, CAD-4 diffractometer; two monitored reflections in each data set examined every 2 h of data collection, examination and 1.7%, respectively); Lorentz, polarization and absorption corrections applied; standard deviations of the structure factors computed as $\sigma(|F_o|) =$ $0.5(VLp)^{1/2}[(Pk + Bg + p^2I_{re1}^2)/I_{re1}]^{1/2}$, where VLp is the standard Lorentz-polarization term, Pk is the peak intensity, Bg is the sum of the background counts taken on both sides of the peak, p is the ignorance factor (0.02 in this work) and I_{re1} is Pk – Bg, all data with $I_{re1} > 3\sigma(I_{re1})$ used in the solution and refinement of the structure.

An origin at the center (2/m) was chosen for the structural refinement; full-matrix least-squares refinements (Larson, 1967), anisotropic thermal parameters, secondary-extinction corrections, reliability values $R_1 = \sum \Delta F / \sum |F_o|$ and $R_2 = [\sum w(\Delta F)^2 / \sum w(F_o)^2]^{1/2}$ where $w = \sigma^{-2}(|F_o|)$ and $\Delta F = ||F_o| - |F_c||$; 'goodness of fit' values (\sum_2) 1.35, 0.95, and 1.47, respectively; atomic scattering factors and anomalous-dispersion correction terms for all atoms from Ibers & Hamilton (1974). Table 1 presents the fractional atomic parameters with e.s.d.'s in parentheses and the equivalent values of the isotropic temperature factors.*

^{*} Lists of structure factors, anisotropic thermal parameters and interatomic distances have been deposited with the British Library Lending Division as Supplementary Publication No. SUP 38104 (8 pp.). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

Table 1. Atomic coordinates $(\times 10^4)$ and equivalent isotropic temperature parameters $(Å^2 \times 10^3)$

E-DO	x	у	Z	U_{eq}^*	
ErPO ₄ Er	0	7500	1250	53	
P O	0	2500 4249 (6)	3750 2165 (8)	52 74	
TmPO ₄ Tm P O	0 0 0	7500 2500 4253 (5)	1250 3750 2155 (5)	57 64 61	
ҮЬРО₄ ҮЬ Р О	0 0 0	7500 2500 4274 (9)	1250 3750 2141 (10)	57 73 86	
$U_{eq} = \frac{1}{3}$ trace \tilde{U} .					

Discussion. A representative stereoscopic drawing of the molecular packing in the unit cell for tetragonal $ErPO_4$, $TmPO_4$, and $YbPO_4$ is shown in Fig. 1, with 50% equiprobability ellipsoids represented. Erbium, thulium, and ytterbium phosphate best fit the spacegroup model $I4_1/amd$. The lanthanoid metal atom is coordinated to eight O atoms forming two unique orthogonal interpenetrating tetrahedra, each of which has unique metal—oxygen bond lengths (see Table 2). The perpendicular-viewed tetrahedron of the bisphenoidal set which is associated with the phosphate group has the shorter discrete M—O distance. The polyhedral structure formed from the O–O contact



Fig. 1. A representative stereoview of the polyhedron in the tetragonal LnPO₄ system.

Table 2. Bond and contact lengths (Å) and angles (°)

M–O –O ⁱ Average	ErPO ₄ 2·297 (4) 2·376 (5) 2·336	TmPO ₄ 2·286 (3) 2·364 (3) 2·325	YbPO ₄ 2·262 (6) 2·357 (6) 2·309
PO	1·531 (4)	1-532 (3)	1·544 (6)
OO ^{II}	2·400 (9)	2-397 (7)	2·419 (12)
OO ^{III,iv}	2·550 (8)	2-554 (6)	2·571 (10)
OP-O [#]	103·2 (5)	102-9 (3)	103·2 (7)
OP-O ^{™,Iv}	112·8 (5)	112-9 (3)	112·7 (7)

Symmetry code: (i) 0, 1-y, -z; (ii) $0, \frac{1}{2}-y, z;$ (iii) $-\frac{1}{4}+y, \frac{1}{4}, \frac{3}{4}-z;$ (iv) $\frac{1}{4}-y, \frac{1}{4}, \frac{3}{4}-z.$

distances (see Table 2) is a distorted dodecahedron (D_{2d}) .

The bond distances given in Table 2 are in good agreement with listed experimental bond distances found in the *Bond Index of the Determination of Inorganic Crystal Structures* (1969–1980). The listed bond and contact lengths found in Table 2 are also internally consistent with the structural model.

The authors thank The Robert A. Welch Foundation in part for financial support (Grant No. AA-668). Further, the authors acknowledge appreciation for the experimental facilities of the Oak Ridge National Laboratory operated by Union Carbide Corporation for the US Department of Energy under contract W-7405-eng-26.

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